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## Short communication

# Facile synthesis of large-area CeO<sub>2</sub>/ZnO nanotube arrays for enhanced photocatalytic hydrogen evolution

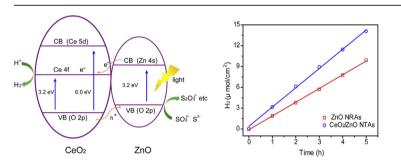


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#### HIGHLIGHTS

- Design of CeO<sub>2</sub>/ZnO nanotube array films via a simple two-step electrodeposition process.
- CeO<sub>2</sub>/ZnO nanotube arrays exhibit enhanced photocatalytic activity and excellent stability.
- Films on FTO substrates make the collection and recycle of photocatalyst easier.

#### G R A P H I C A L A B S T R A C T



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#### ABSTRACT

 $CeO_2/ZnO$  nanotube array films grown on FTO substrates are successfully synthesized by a two-step electrodeposition process. The  $CeO_2/ZnO$  nanotube array films exhibit substantially enhanced photocatalytic activity with a hydrogen evolution rate of  $2.7~\mu mol~cm^{-2}~h^{-1}$  under white light irradiation and excellent stability. Moreover, the collection and recycle of these immobilized  $CeO_2/ZnO$  nanotube array films are much easier compared with the powder photocatalysts.

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#### 1. Introduction

As a clean energy carrier without greenhouse gases or other pollutants, hydrogen has gained great interest owing to the increasing demand for energy and environment protection. Searching an effective and low-cost approach to produce hydrogen

is very important for hydrogen economy. Photocatalytic hydrogen production is widely recognized as one of the most promising approaches to generate hydrogen, which is produced in a sustainable manner from solar energy without yielding carbon dioxide [1,2]. Recently, nanostructures have received considerable interest for photocatalytic hydrogen production due to their high surface area and rapid charge separation ability. Among various nanostructures, hollow nanotubes are attracting a great deal of attention as photocatalysts because they have higher interfacial area and faster charge transport pathway than nanoparticles and other one-

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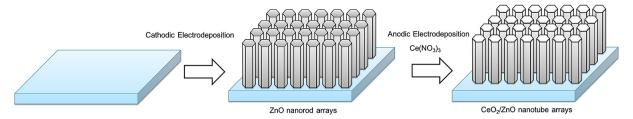


Fig. 1. Schematic diagram of the fabrication process for CeO<sub>2</sub>/ZnO nanotube arrays.

dimensional (1D) nanostructures such as nanorods or nanowires [3,4]. Moreover, recent reports have demonstrated that the direct growth of ordered nanotubes on conductive substrates can facilitate the diffusion of active species and transport of electrons, and could further improve photocatalytic performance [5,6]. In recent years, numerous methods including high temperature evaporation, hydrothermal synthesis, anodic oxidation, and template- and surfactant-assisted growth techniques have been developed to prepare nanotubes on conductive substrates [7–11]. Nevertheless, the development of facile, low-cost and effective methods for the large-area synthesis of ordered nanotubes on conductive substrates is highly desirable, but still challenging.

Transition metal oxides such as TiO<sub>2</sub>, ZnO and BiVO<sub>4</sub> have been extensively investigated as photocatalysts for photocatalytic hydrogen production [12–17]. Little attention has been paid to the rare earth oxides, regardless of their outstanding optical and catalytic properties. In recent years, CeO<sub>2</sub> has attracted increasing interest as a photoactive material because it has strong redox capability, and a relatively small band gap of 3.2 eV that originates from the O 2p to Ce 4f transition [18]. Additionally, CeO2 nanostructures such as hexagonal nanorods and nanowires have been examined in photocatalytic hydrogen evolution, and they have shown good photocatalytic activity [6,19,20]. However, the largescale fabrication of CeO<sub>2</sub> nanostructures with good photocatalytic activity, especially for commercial manufacture, is still very challenging. In this paper, we report a facile and cost-effective electrochemical method to synthesize large-area CeO<sub>2</sub>/ZnO nanotube arrays (NTAs) on F-doped SnO2 (FTO) glass substrates and their improved performance in photocatalytic hydrogen production. The CeO<sub>2</sub>/ZnO nanotube arrays were grown on FTO substrates via a two-step electrodeposition process, which involves the cathodic electrodeposition of aligned ZnO nanorod arrays (NRAs) as the template and subsequently anodic electrodeposition of CeO<sub>2</sub> on the ZnO nanorods to form CeO<sub>2</sub>/ZnO nanotubes. These CeO<sub>2</sub>/ZnO nanotube arrays grown directly on FTO substrates can provide a high interfacial area and superior electron transport pathways, which hence improve their photocatalytic performance. The CeO<sub>2</sub>/ ZnO nanotube arrays exhibited substantially higher photocatalytic activity than ZnO nanorod arrays, with a hydrogen evolution rate of  $2.7~\mu mol~cm^{-2}~h^{-1}$  under white light irradiation and good cycling performance.

## 2. Experimental

#### 2.1. Materials

The analytical grade of commercially Zn(NO<sub>3</sub>)<sub>2</sub> (Sinopharm Chemical Reagent Co., Ltd., Beijing, China), Ce(NO<sub>3</sub>)<sub>3</sub> (Sinopharm Chemical Reagent Co., Ltd., Beijing, China), ammonium acetate (NH<sub>4</sub>Ac) (Sinopharm Chemical Reagent Co., Ltd., Beijing, China), hexamethylenetetramine (HMT) (Sinopharm Chemical Reagent Co., Ltd., Beijing, China) and dimethyl sulfoxide (Sinopharm Chemical Reagent Co., Ltd., Beijing, China) were used directly without any

purification. The F-doped  $SnO_2$ -coated glass (FTO, TCO-15) with a sheet resistance of 14 <0MEGA> per square was purchased from Nippon Sheet Glass Co., Ltd., Tokyo, Japan. The FTO glass was cleaned ultrasonically in distilled water, ethanol, and acetone and then rinsed in distilled water again before electrodeposition.

#### 2.2. Photocatalysts preparation

The CeO<sub>2</sub>/ZnO nanotube arrays were synthesized by a two-step electrodeposition process, which is summarized schematically in Fig. 1. First, ZnO nanorod arrays were grown on FTO substrates (4  $\times$  4 cm²) by cathodic electrodeposition in an aqueous solution containing 0.02 M Zn(NO<sub>3</sub>)<sub>2</sub>, 0.01 M NH<sub>4</sub>Ac and 0.01 M HMT with a current density of 0.5 mA cm $^{-2}$  at 90 °C. After 60 min cathodic deposition, a white film was obtained on FTO substrate. The ZnO nanorod arrays were then used as templates for further formation of CeO<sub>2</sub>/ZnO nanotube arrays. CeO<sub>2</sub>/ZnO nanotubes were directly obtained via anodic electrodeposition in a solution of 0.01 M Ce(NO<sub>3</sub>)<sub>3</sub> + 10% DMSO (10 vol % DMSO:90 vol % H<sub>2</sub>O) with a current density of 0.25 mA cm $^{-2}$  at 90 °C.

#### 2.3. Characterizations and measurements

The surface morphologies and microstructures were analyzed by a field-emission scanning electron microscope (FE-SEM, JSM-6330F) and transmission electron microscopy (TEM, JEM-2010F). The structures of the samples were investigated via an X-ray diffractometer (XRD, D8 ADVANCE) using Cu K $\alpha$  radiation ( $\lambda=0.15418$  nm). The chemical state of the films was analyzed directly using X-ray photoelectron spectroscopy (XPS, ESCA-Lab250). The optical properties of the products were measured with a UV–Vis–NIR Spectrophotometer (UV–Vis–NIR, Shimadzu UV-2450).

The photocatalytic hydrogen evolution reactions were examined in a closed gas circulation and evacuation system. Typically, a piece of large-area ZnO or  $CeO_2/ZnO$  films (4  $\times$  4 cm²) was immersed into 100 mL of 0.1 M Na<sub>2</sub>S and 0.1 M Na<sub>2</sub>SO<sub>3</sub> mixed aqueous solution in a Pyrex reaction cell. The light source was a 300 W Xe lamp (PLS-SXE-300UV, Beijing Trusttech) supplying the full wavelength illumination. The intensity of the white light was determined by an irradiatometer (BNU photoelectric Co., Ltd., FZ-A) and was about 100 mW cm². The amount of hydrogen produced was analyzed using on-line gas chromatography (with a thermal conductivity detector and an N<sub>2</sub> carrier).

### 3. Results and discussion

Fig. 2a is the photo image of CeO<sub>2</sub>/ZnO nanotube arrays obtained on FTO substrate, showing that most of the FTO substrate (around 16 cm<sup>2</sup>) was uniformly covered by a thin film with good transmittance. Scanning electron microscopy (SEM) images in Fig. 2b and c clearly show that the white film obtained on FTO substrate was covered by dense and ordered ZnO nanorods. These nanorods

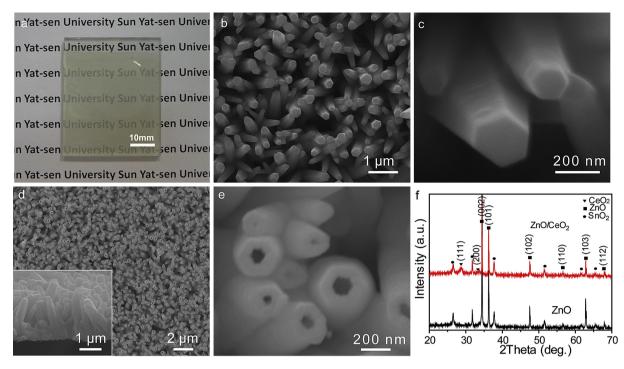


Fig. 2. (a) Photo image of CeO<sub>2</sub>/ZnO NTA film on FTO substrate. (b, c) SEM images of ZnO NRAs. (d, e) SEM of typical CeO<sub>2</sub>/ZnO NTAs. (f) XRD patterns of the ZnO NRAs and CeO<sub>2</sub>/ZnO NTAs.

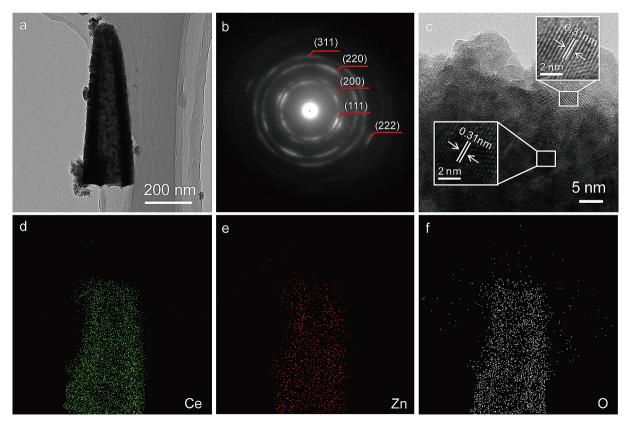


Fig. 3. (a) Conventional, (b) the corresponding SAED pattern obtained at the tip of the  $CeO_2/ZnO$  NTA film (c) lattice-resolved TEM images of  $CeO_2/ZnO$  NTA film and (d-f) EDX elemental mapping images of the Ce, Zn and Qn, respectively.

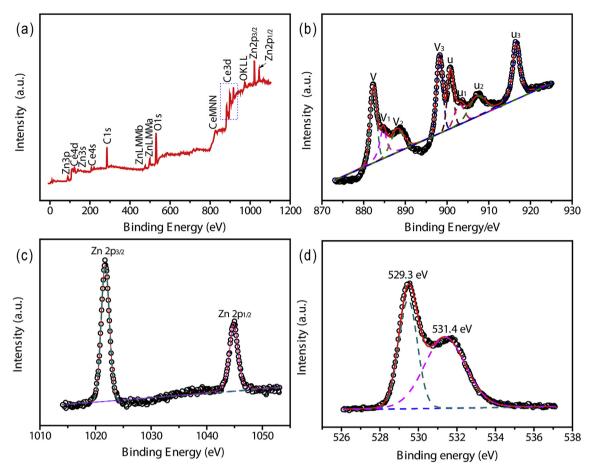


Fig. 4. (a) XPS survey, (b) Ce 3d, (c) Zn 2p and (d) O 1s core level spectra collected for the CeO<sub>2</sub>/ZnO NTAs.

have a diameter of 150–300 nm. This clearly demonstrates that this present method can produce a good quality thin film on a large scale. Fig. 2d and e show the SEM images of CeO<sub>2</sub>/ZnO nanotube arrays, indicating that hollow nanotubes were successfully formed after the anodic deposition. The diameter of these nanotubes was about 150-350 nm, which is similar to that of the ZnO nanorod precursors. XRD spectra collected from ZnO nanorod arrays and CeO<sub>2</sub>/ZnO nanotubes are shown in Fig. 2f. All the diffraction peaks of ZnO nanorod arrays can be well indexed as the hexagonal wurtzite structure of ZnO (JCPDS # 36-1451). After anodic deposition in Ce(NO<sub>3</sub>)<sub>3</sub> solution, clear diffraction peaks corresponding to cubic CeO<sub>2</sub> (JCPDS # 65-2975) were observed and the intensity of ZnO peaks decreased, indicating CeO<sub>2</sub> was formed on the ZnO. Compared with other nanotube arrays obtained via ZnO nanorod arrays as templates, the present method is a simplification because the ZnO template is directly removed during the process of anodic electrodeposition of CeO<sub>2</sub>. The resulting large void volume provides a three-dimensional (3D) space for the mass transport of reactant and product molecules.

More detailed microstructural information of the  $CeO_2/ZnO$  nanotube films was further investigated by transmission electron microscopy (TEM). Fig. 3a shows the typical image of single  $CeO_2/ZnO$  nanotube with the diameter of  $\sim 200$  nm, implying that the nanotube was successfully synthesized. A typical selected area electron diffraction (SAED) pattern of a  $CeO_2/ZnO$  nanotube was presented in Fig. 3b, which clearly demonstrates the  $CeO_2/ZnO$  nanotube is polycrystalline and  $CeO_2$  is dominated in the prepared sample. Fig. 3c is the high-resolution TEM (HRTEM) image of a  $CeO_2/ZnO$  nanotube, disclosing that the  $CeO_2$  layer is highly

crystalline. The clear lattice fringe with a d-spacing of 0.31 nm belongs to the lattice fringe of the (111) plane of the CeO<sub>2</sub>. Energy dispersive X-ray spectroscopy (EDX) analysis was carried out to analyze the composition of the nanotube. The elemental mapping images of this single nanotube are shown in Fig. 3d–f, clearly indicating that Zn, Ce and O are uniformly distributed throughout the entire nanotube.

In order to further study the composition of CeO<sub>2</sub>/ZnO NTAs, Xray photoelectron spectroscopy (XPS) was also conducted. Fig. 4a displays the XPS survey of CeO2/ZnO NTA films. Only C, Zn and O signals are observed (C single comes from the adventitious carbon), which reveals the high purity of the sample. Fig. 4b displays the Ce 3d core level spectrum. The peaks labeled as  $v_1$  and  $u_1$  are the characteristic peaks of Ce<sup>3+</sup> states while others are the characteristic peaks of  $Ce^{4+}$  states [21]. This indicates most of Ce ions are present in the form of  $Ce^{4+}$  states. The Zn 2p core level spectrum (Fig. 4c) shows that the Zn  $2p_{3/2}$  and Zn  $2p_{1/2}$  are located at binding energies of 1021.0 eV and 1044.0 eV, which are consistent with the values reported recently for ZnO [22]. The O 1s core level spectrum can be deconvolved into two peaks (Fig. 4d). The peak at 529.3 eV is a characteristic value for oxygen located in the metal oxide crystal lattice and the defect oxygen or hydroxyl groups, respectively [23]. These results clearly indicate that the prepared product is CeO<sub>2</sub>/ ZnO NTAs, which is in accordance with the TEM and EDX results.

Fig. 5 compares the UV—Vis absorption spectra of ZnO NRA and  $CeO_2/ZnO$  NTA films. The absorption band edge of  $CeO_2/ZnO$  NTA film shows a slightly red shift compared to the pure ZnO NRA film. Recent reports have shown that the transformation from  $Ce^{4+}$  to  $Ce^{3+}$  may lead to a diversification of its optical properties [24]. As

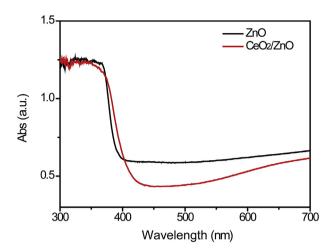
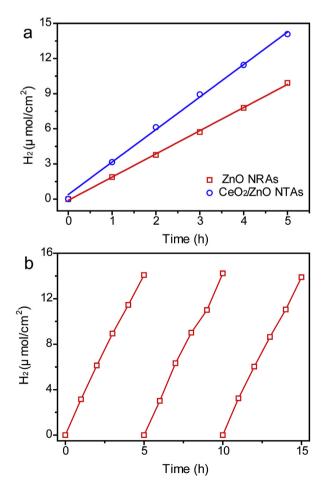


Fig. 5. UV-Vis absorption spectra of ZnO NRA film and CeO<sub>2</sub>/ZnO NTA film.

demonstrated by XPS results (Fig. 4b), a small amount of  $Ce^{3+}$  is presented in as-prepared  $CeO_2/ZnO$  NTAs. Therefore, it may be reasonably concluded that the red-shift of the band edge results from the  $Ce^{4+}$  to  $Ce^{3+}$  transformation. This is beneficial for harvesting light during the photocatalytic process and giving high quantum efficiency for hydrogen production [6,20,25].



**Fig. 6.** (a) Photocatalytic hydrogen production rate collected for ZnO NRA film,  $CeO_2/ZnO$  NTA film in a solution containing 0.1 M Na<sub>2</sub>S and 0.1 M Na<sub>2</sub>SO<sub>3</sub> under 100 mW cm<sup>-2</sup> white light irradiation. (b) Cycling performance of  $CeO_2/ZnO$  NTA film.

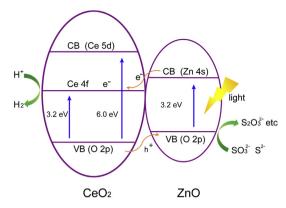


Fig. 7. Photocatalytic mechanism of hydrogen evolution on the CeO<sub>2</sub>/ZnO NRA film.

To evaluate the photocatalytic activity of the CeO<sub>2</sub>/ZnO NTA film, the rate of the hydrogen evolution was examined in a mixed 0.1 M Na<sub>2</sub>S and 0.1 M Na<sub>2</sub>SO<sub>3</sub> solution under 100 mW cm<sup>-2</sup> white light irradiation. The time dependent rate of hydrogen evolution is showed in Fig. 6a. The ZnO NRA film has an average hydrogen evolution rate of 1.9  $\mu$ mol cm<sup>-2</sup> h<sup>-1</sup>. Significantly, the CeO<sub>2</sub>/ZnO NTA film achieved about 2.7  $\mu$ mol cm<sup>-2</sup> h<sup>-1</sup>, which is nearly 40% higher than the ZnO NRA film. The efficiency of the solar to hydrogen ( $\eta$ ) over CeO<sub>2</sub>/ZnO nanotube array film was determined by comparing the heat of combustion of hydrogen evolved to the incoming luminous energy, and was calculated using the equation:

$$\eta = E_{\rm H_2}/E_{\rm solar} = [\Delta H \cdot \nu(\rm H_2) \cdot t \cdot S]/(Ee \cdot t \cdot S) = \Delta H \cdot \nu(\rm H_2)/Ee$$
(1)

where the  $\Delta H$  is the standard free enthalpy change  $(H_2 + 1)$  $2O_2 \rightarrow H_2O$ , -241.8 kJ mol<sup>-1</sup>),  $\nu(H_2)$  is average hydrogen evolution rate (2.7  $\mu$ mol cm<sup>-2</sup> h<sup>-1</sup>) and *Ee* is the intensity of white light irradiation (100 mW cm<sup>-2</sup>). The calculated CeO<sub>2</sub>/ZnO nanotube array film achieves an energy conversion efficiency of 1.8%. Noteworthy is that this calculation does not consider the cathodic regeneration of the complex redox system, and the necessary anodic evolution of oxygen to complete the process. The enhanced photocatalytic activity for CeO<sub>2</sub>/ZnO NTA film may be attributed to the following factors. First, the nanotubes have larger surface area and more active sites than nanorods. The unique nanotube structure not only allows the active species diffuse into the inside of the nanotubes, but also allows more active sites to participate in the photocatalytic reaction, hence improving the performance. Second, CeO<sub>2</sub> has a strong redox capability, which is an efficient catalytic material for hydrogen evolution, as recent reports have indicated [6,20]. Finally, CeO<sub>2</sub> can form type-II structures with ZnO, as shown in Fig. 7. Under irradiation, the photoexcited electrons produced at the conduction band of ZnO will be first transfer into the Ce 4f orbital rather than recombining with holes in the valence band of ZnO. Meanwhile, the photoexcited holes will flow from the valence band of CeO<sub>2</sub> to the valence band of ZnO. As a result, the formation of type-II structure can effectively suppress charge recombination, thus improving photocatalytic activity. Besides the high hydrogen evolution activity, long-term stability is another indispensable factor for a high-performance of the photocatalyst. Fig. 6b shows the rate of photocatalytic hydrogen evolution on the CeO<sub>2</sub>/ZnO NTA film as a function of time to evaluate its stability. Hydrogen production is proportional to illumination time during the entire measurement interval, suggesting that no performance decay occurs.

Significantly, the  $CeO_2/ZnO$  NTA films developed in this work have another advantage over traditional photocatalysts, which are

usually powders, and are difficult to collect and recycle without loss, even resulting in environmental pollution. In contrast, the collection and recycling of  $CeO_2/ZnO$  NTA film in this case are very simple, since they can be removed from the cell and washed with deionized water, eliminating mass loss and environmental pollution.

#### 4. Conclusions

In summary, we have demonstrated the design and synthesis of  $CeO_2/ZnO$  NTA films on FTO-glass substrates. These highly ordered  $CeO_2/ZnO$  NTA films with large specific area were synthesized by a two-step electrodeposition process. The photocatalytic hydrogen rate of  $CeO_2/ZnO$  NTA film reaches 2.7  $\mu$ mol cm $^{-2}$  h $^{-1}$ , which is substantially higher than that of the corresponding ZnO NRA film. The enhancement in photocatalytic performance may be attributed to large surface areas, more reaction sites and the type-II structures which can reduce the recombination of the photoexcited electron—hole pairs. Additionally, the  $CeO_2/ZnO$  NTA films grown the FTO-glass substances show much better stability and recyclability as compared to powder photocatalysts. These findings are of great significance for the design of effective photocatalysts.

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